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An nth-order, Gaussian Energy Distribution Model for Sintering

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Abstract

Although it is well known that the rate of sintering is governed by deceleratory kinetics, it

is often difficult to fit power-law and nth-order reaction models over broad time-

temperature ranges. This work shows that a phenomenological model combining a

reaction order with an activation energy distribution can correlate surface area as a

function of sintering time and temperature over a greater range of those variables.

Qualitatively, the activation energy distribution accounts the dependence of free energy

on particle size and material defects, while the reaction order accounts for geometric

factors such as a distribution of diffusion lengths. The model is demonstrated for

sintering of hydroxyapatite using data of Bailliez and Nzihou (Chem. Eng. J. 98 (2004),

141-152).

Keywords: sintering, kinetics, activation energy distributions, modeling, hydroxyapatite

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1. Introduction

Sintering of powders is an industrial practice spanning many applications, and German¹ gives an excellent introduction. In very general terms, sintering is governed by two interrelated properties, a change in free energy, which provides the driving force for the process, and kinetics, which provides the mobility of the system to the lower free energy state. Sintering can occur by many mechanisms, including viscous flow, plastic flow, evaporation-condensation, surface diffusion, volume diffusion, and grain-boundary diffusion.

Predicting the kinetics of sintering has practical aspects for both process optimization and material lifetime prediction. Process optimization is usually easier, because it is usually an interpolation problem for which the calibration data might cover a relatively narrow range of conditions. Lifetime prediction is more challenging, since it involves extrapolation of artificial aging experiments outside the range of calibration. Consequently, a relatively small deviation in a model at the extremes of the calibration data can result in a relatively large error in lifetime prediction if the functional form is not correct.

A vast literature exists on the kinetics of sintering, and various equations have been derived that use powers of time and particle size along with an Arrhenius temperature dependence. However, these models often have difficulty correlating sintering data over wide ranges of time and temperature. For example, the common nth-order sintering model often requires n to be a function of temperature, with the qualitative

justification that the mechanism is changing as a function of temperature (e.g., Bailliez and Nzihou²).

Changes in free energy driving the sintering process are ordinarily attributed to changes in the radius of curvature. Less widely recognized is that sintering often starts with very imperfect crystals that have free energies substantially different from the perfect material. For example, Rogers and Dinegar³ report that heats of fusion of pentaerythritol tetranitrate (PETN) crystals can have heats of fusion up to 20% less than the single crystal value, and the variation of the heat with crystallization conditions is far greater than with changes in surface area caused by grinding. A free energy distribution in the starting material will result in an activation energy distribution in the kinetics.

Both nth-order and activation energy distribution kinetic models have been used extensively for modeling fossil fuel conversion. 4,5,6,7 The earliest and simplest energy distribution model used a Gaussian distribution characterized by a mean, E_0 , and standard deviation, σ . For systems with modest distributions of reactivity, nth-order ($n \le 2.5$) or Gaussian ($\sigma \le 3.5\%$ of E_0) models work equally well, even with considerable extrapolation in temperature. However, sintering is often, if not usually, characterized by reaction orders that are considerably larger, and possibly by free energy distributions as well, if the Rogers-Dinegar result for PETN is typical.

The hypothesis tested in this paper is that the temperature dependence of the reaction order can be removed if the kinetic model also includes an activation energy distribution. We use the data of Bailliez and Nzihou,² since it covers such a broad range of temperature and degree of sintering. We find that the *n*th-order/Gaussian distribution

model works very well, resulting in a reduction in nonlinear-regression residuals compared to the nth-order model typically used for sintering.

2. Sintering Models

Deceleratory sintering reactions are often characterized by a power law in time^{9,10}

$$\alpha \propto (1+at)^{-\nu} \tag{1}$$

where α is the extent of reaction (e.g., ratio of the change in surface area to the ultimate change in surface area, or 1- S/S_0) and a and v are constants, and t is time; or an nth-order reaction, 2,11,12

$$-d\alpha/dt \propto (1-\alpha)^n \tag{2}$$

Coming from different fields, Raynaud et al.¹³ and Tarutis¹⁴ independently note that the two approaches are actually equivalent, with the exponent of the power law in time being related to the order of the nth-order reaction by n = 1 + 1/v. Reaction order is commonly interpreted in geometric terms, e.g., shrinking-core reactions are described by n < 1.¹⁵ Tarutis, drawing upon earlier work by Boudreau and Ruddick, ¹⁶ notes that an nth-order reaction is mathematically equivalent to for n > 1 to a Gamma (near-exponential) distribution of reactivity. Consequently, one can consider reaction order as a measure of a distribution of diffusion lengths, for example. Regardless of the precise physical

interpretation, n should be constant if the geometric progression of the reaction is independent of temperature, and all temperature dependence would be ascribed to a single activation energy, if one uses the standard Arrhenius rate law.

Alternatively, one can use a Gaussian distribution of activation energies to describe the distribution of reactivity. In this case, ¹⁷

$$d\alpha/dt = \int_0^\infty k(E) \exp[-\int_0^t k(E)dt]D(E) dE \qquad (1^{st}\text{-order})$$
 (3)

and

$$d\alpha/dt = \int_0^\infty k(E) \left[1 - (1 - n) \int_0^t k(E) dt\right] D(E) dE$$
 (nth-order) (4)

where $k = A \exp(-E/RT)$, E is the activation energy, A is the frequency factor, R is the gas constant, and

$$D(E) = (2\pi)^{-1/2} \sigma^{-1} \exp[-(E - E_0)^2 / 2\sigma^2]$$
 (5)

where E_0 is the mean energy and σ is the standard deviation. In practice, equation is implemented in the LLNL kinetics analysis program (Kinetics98,⁴ Kinetics2000,¹⁸ and Kinetics05¹⁹) by discretizing the distribution into parallel independent 1st- or nth-order reactions having a maximum spacing of 2.1 kJ/mol to cover the required energy interval and then weighting these reactions according to the Gaussian distribution. Each reaction

is numerically integrated using the rational approximation to the exponential integral given by Braun and Burnham.¹⁷

Both the simple nth-order and 1^{st} -order Gaussian distribution approaches yield deceleratory curves at constant temperature, with the ultimate extent of reaction appearing to depend on temperature when n and σ are large. Calculations for intermediate values of both n and σ are given in Figure 1. In order to have the overall degree of sintering cover the same range, a higher mean energy is needed for the Gaussian model so that the lowest energy channel of the distribution is close to the single value of the nth-order reaction. The shape of the deceleration and how it varies with temperature is different for the two models.

Sintering of hydroxyapatite

Bailliez and Nzihou² provide an interesting data set for testing the ability of a combined nth-order activation energy distribution model to correlate the extent of sintering over a very wide range. They present data for two hydroxyapatites: HAP_{TCP} was formed by reacting $CaCl_2$, H_3PO_4 , and NaOH; and HAP_{CaO} was formed by reacting $Ca(NO_3)_2$ with $(NH_4)_2HPO_4$ and ammonia. The initial surface areas were 28 and 104 m^2/g , respectively.

Data was digitized from the published plots of surface area versus time at various temperatures. It was then fitted by nonlinear regression to nth-order, Gaussian, and combined models using the LLNL analysis program Kinetics05. The parameters $\ln(A)$, E/10000R, n, and σ are optimized to a relative tolerance of 0.001. Results of this analysis

are given in Table 1, including a measure of the uncertainties in E, n, and σ . A graphical comparison of data with calculation is given in Figure 2 for the combined model.

When only one of the two parameters (n or σ) is used, a better fit is obtained with the nth-order model for HAP_{TCP} and with the Gaussian model for HAP_{CaO}. However, the best fit is obtained for both materials using both model parameters, and it is especially better for HAP_{TCP}. The mean activation energy for the Gaussian model is higher than for the nth-order model as explained in the previous section. In work to be reported elsewhere, the energy distribution parameter (σ) is much more important than reaction order (n) for correlating sintering data of the energetic material PETN— negligible improvement in the RSS is obtained by optimizing n in addition to σ , and the optimized value of n is close to unity. This result suggests that using both n and σ for HAP is indeed meaningful.

Two related issues are the uncertainties in the parameters and when one should use four rather than three parameters to describe the data. Unlike for linear regression, such questions cannot be answered by definitive mathematical formulas. Although linearization methods are commonly used to estimate parameter uncertainties in nonlinear regression, they may be grossly in error unless the model is close to linear. Consequently, the uncertainties reported in Table 1 are those that change the RSS by 10% for the parameter of interest while re-optimizing the other parameters. This difference is large compared to the convergence criteria but comparable to the improvement by using both n and σ instead of the best one of the two. The relative changes in $\ln(A)$ would be similar to those in E/R, since those variables are highly correlated per the well known compensation law. Of most significance is that the uncertainties in n and σ when both

are used are small compared to their values, which supports the conclusion of the previous paragraph.

The activation energies reported here are about one-third lower than those reported by Bailliez and Nzihou². The reason for this discrepancy is not certain, but it may relate to the common problem of deriving activation energies under circumstances where the extent of conversion is not maintained constant. In this case, the activation energy can shift from its true value to make up for other model deficiencies.²¹ This is not possible by our method, since the entire data set is fitted simultaneously to the model. Even so, the activation energies are not exceptionally well constrained by the data even in our case, as indicated by the uncertainties given in Table 1.

Conclusions

The combination of an *n*th-order reaction model with a Gaussian activation energy distribution provides a simple yet powerful method for correlating sintering data over a very wide time-temperature range. The *n*th-order aspect can be interpreted in terms of standard neck-growth phenomena and a distribution of diffusion lengths. The activation energy distribution reflects the distribution of free energies for imperfect starting crystals. While the combined model fits the data better than either aspect independently for both hydroxyapatite samples examined, the activation energy distribution aspect is considerably more important for the higher surface area material. The activation energies derived by nonlinear regression to the entire range of conversion

simultaneously are more reliable than other forms of model fitting that sample different aspects of the reaction at different temperatures.

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Table 1. Hydroxyapatite kinetic parameters derived by nonlinear regression for the Gaussian and nth-order reaction models from the data of Bailliez and Nzihou².

	A, s^{-1}	E_0/R , K	n	σ , % of E_{θ}	RSS ^a
HAP _{TCP}					
nth-order	8.43×10 ¹¹	30313	6.89	0.0	0.1001
1.1 RSS ^b		±7684	±1.56	_	
Gaussian	1.63×10 ¹¹	32506	1.00	17.17	0.3090
1.1 RSS		±16311	_	±2.85	_
Both	1.64×10 ¹³	33001	7.01	6.93	0.0762
1.1 RSS	_	±7160	±0.32	±1.47	_
HAP _{CaO}					
nth-order	1.79×10 ⁹	24246	4.01	0.0	0.1530
1.1 RSS	_	±3585	±0.73		_
Gaussian	9.44×10 ⁹	27383	1.00	11.4	0.1189
1.1 RSS		±4826	—	±1.16	
Both	2.92×10 ¹⁰	27149	3.22	8.32	0.1046
1.1 RSS	_	±4431	±0.89	±1.40	_

^aresidual sum of squares from nonlinear regression.

^bchange in parameter to increase RSS by 10% while re-optimizing other variables.

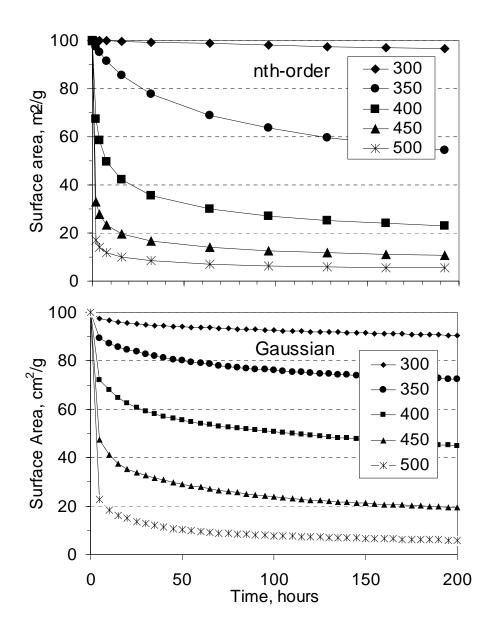


Fig. 1. Idealized sintering curves calculated from nth-order (top) and Gaussian activation energy distribution (bottom) models. In both cases, $A=3\times10^{15}$ s⁻¹. For the *n*th-order model, n=5 and E/R=30,000 K. For the Gaussian model, $\sigma=10\%$ of E_0 and $E_0=33,000$ K. The higher mean energy is needed for the Gaussian model so that the lowest energy channel of the distribution is close to 30,000 K.

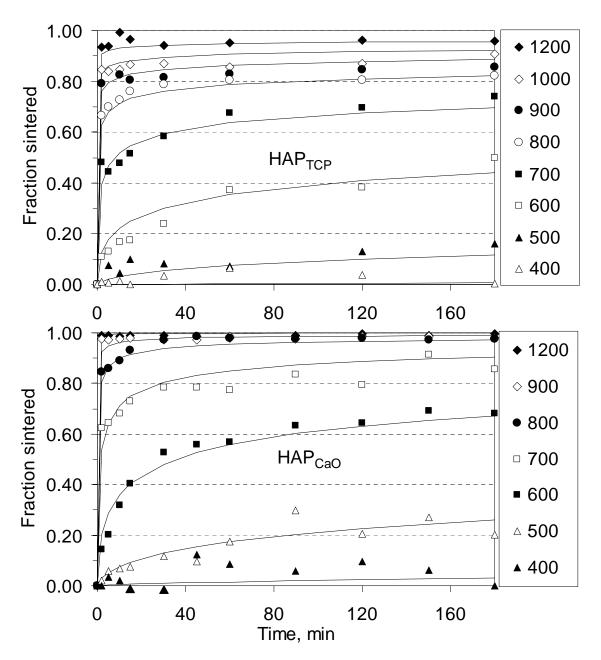


Figure 2. Comparison of measured and calculated fractions sintered $(1-S/S_0)$ for the nth-order Gaussian energy distribution model. The four model parameters are fitted simultaneously by nonlinear regression and are given in Table 1.

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